# Raman Spectroscopy Vibrational Analysis of Octahedrally Coordinated Fluorides: Application to Transition Metal Fluoride Glasses

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An extensive Raman investigation of octahedrally coordinated single-crystal fluorides have been performed. The theoretical study of these phases, using group theory, gives the evidence of close relationship between metal oxidation number, connection schemes, and spectra features. The results of this study, when applied to the case of transition metal fluoride glasses, propose these glasses to be built up with octahedral entities predominantly linked by corners to form one-dimensional networks. © 1989 Academic Press, Inc.

# Introduction

Fluoride glasses belong to a new class of materials with high potentiality for fiber optics because of their ultra transparency in the middle infrared and a very low estimated loss value (about  $10^{-3}$  dB/km at 2.55  $\mu$ m) (1, 2).

Three-dimensional (3d) transition metal fluoride glasses (TMFG) have been extensively investigated in various fields (X-ray and neutron diffraction (3-5); EXAFS (6, 7), NMR (8), magnetism (9)) to get coherent crystal chemistry knowledge.

Raman spectroscopy seems to be another way to access the structure of these materials. With a large number of single crystals of octahedrally coordinated 3d transition metals in various known structural types we looked at correlations between the vibration modes and the oxidation number and/or the connection scheme. These data were used to interpret the Raman spectra of the fluoride glasses (TMFG) which contain the same entities.

#### **I. Experimental Procedures**

#### 1. Sample Preparations

Crystalline samples were obtained either from a chloride flux method (10) (Na<sub>5</sub>Al<sub>3</sub>  $F_{14}$ , Cs<sub>2</sub>NaAl<sub>3</sub> $F_{12}$ , Rb<sub>2</sub>CrF<sub>5</sub>, CoF<sub>2</sub>, MnF<sub>2</sub>, NaMnCrF<sub>6</sub>,  $\beta$ -AlF<sub>3</sub>, K<sub>2</sub>NaGaF<sub>6</sub>) or HF hydrothermal synthesis (11) ((NH<sub>4</sub>)<sub>3</sub>AlF<sub>6</sub>, Tl<sub>2</sub> AlF<sub>5</sub> · H<sub>2</sub>O, TlAlF<sub>4</sub>, KAlF<sub>4</sub>,  $\beta$ -RbAlF<sub>4</sub>).

Bulk glass samples are prepared as already described (12) (by melting the mixture of anhydrous fluorides in a platinum crucible; the melt is quenched on a preheated brass mold); they are then polished with  $OH^-$  free solvent.

# 2. Raman Measurements

Raman spectra were recorded with a DILOR Z24 spectrometer (triple monochromator, photon counting electronics); 514.5- and 457.9-nm lines of an Argon-ion laser (Coherent INOVA 90) have been used as the excitation source.

Single-crystal Raman spectra were recorded under a microscope (ULF  $\times$ 50 objective) from 20 to 600 cm<sup>-1</sup> with 700- $\mu$ m slit width. Although imperfectly defined under the microscope, polarization effects may be qualitatively taken into account. The intensities of all the Raman spectra are given with arbitrary units; a typical polarized Raman spectrum is shown in Fig. 1.

The measurements for TMFG were carried out on large polished samples in the 90° scattering configuration in both the polarized H–H configuration (the scattered light is analyzed for electric vector parallel to that of the incident radiation) and the H-Vconfiguration (electric vector perpendicular to incident light).

# II. Vibrational Study of Crystalline Fluorides

# 1. Crystal Structures of Studied Phases

The most important features of their crystal chemistry are described in the following lines.

(a) Isolated octahedra. The Fm3m cubic structure of  $(NH_4)_3AlF_6$  (13) or elpasolite,  $K_2NaGaF_6$  (14), is shown in Fig. 2.

(b) One-dimensional network structures. Two examples of structures built up from isolated chains of corner-shared octahedra  $(Tl_2AIF_5 \cdot H_2O \ (15) \ trans-chains and Rb_2 \ CrF_5 \ (16) \ cis-chains)$  are shown in Figs. 3 and 4, respectively.

(c) Two-dimensional network structures. In the fluoroaluminate case, we can distinguish two kinds of  $(AlF_4)_n$  layers:



FIG. 1. Polarized Raman spectrum of Rb<sub>2</sub>CrF<sub>5</sub>.



FIG. 2. Fm3m structure of (NH<sub>4</sub>)<sub>3</sub>AlF<sub>6</sub>.

-tetragonal as in TlAlF<sub>4</sub> (17) (Fig. 5) with its isotypes KAlF<sub>4</sub> (17) and  $\beta$ -RbAlF<sub>4</sub> (18), or in Na<sub>5</sub>Al<sub>3</sub>F<sub>14</sub> (19) (Fig. 6);

—hexagonal as in hexagonal tungsten bronze (H.T.B.)-type  $Cs_2NaAl_3F_{12}$  (20) (Fig. 7).

(d) Tridimensional structures. Various structural types were studied:

—The rutile structure has edge-shared chains of octahedra linked together by corners as in  $CoF_2$ ,  $MnF_2$  (21) (Fig. 8).

—The H.T.B.-type  $\beta$ -AlF<sub>3</sub> (22) is shown in Fig. 9.



FIG. 3. Structure of  $Rb_2AlF_5 \cdot H_2O$ . The octahedra are *trans*-connected.  $\bigcirc$ ,  $H_2O$  molecules.



FIG. 4. Structure of  $Rb_2CrF_5$ ; the octahedra are *cis*-connected.

—The NaMnCrF<sub>6</sub> (23) structure (Fig. 10) has octahedra linked by corners and by edges with two different oxidation number cations ( $Mn^{2+}$  and  $Cr^{3+}$ ).

# 2. Discussion

(1) Evidence of pure symmetric stretching (SS). Raman spectroscopic studies performed on  $AAIF_4$  (A = K, Rb, Cs, NH<sub>4</sub>) compounds show that the  $A_{1g}$  band (VV polarized) of the Raman active vibrations of  $AIF_{6}^{-}$  octahedron lies in the range



FIG. 5. Structure of TlAlF<sub>4</sub>.



FIG. 6. Structure of  $Na_3Al_3F_{14}$  showing  $[Al_3F_{14}]_n^{5n-1}$  sheets.

515–545 cm<sup>-1</sup>. Lattice dynamics calculations unambiguously show that this band is assigned to the symmetric stretching (SS) of the nonbridging F atoms  $F_{nb}$  (24, 25); it is



FIG. 7. Structure of  $Cs_2NaAl_3F_{12}$ ; Al atoms form triangular and hexagonal rings.



FIG. 8. Rutile structure.

the case for crystals with isolated octahedra or mono and bidimensional networks.

Figure 11 shows the Raman spectra obtained for  $K_2NaGaF_6$ ,  $Tl_2AlF_5 \cdot H_2O$ , and  $Cs_2NaAl_3F_{12}$ .

According to Rousseau *et al.* (26), when neglecting F-F interactions, the harmonic force constant associated with the motion of F atoms in the M-F direction of  $MF_6$ octahedron may be written in the form

$$K_{\rm F\parallel} = \frac{e^2}{16r^3} (2A_2 - \alpha Z_{\rm M} Z_{\rm F}),$$
 (1)

where  $Z_{\rm M}$  and  $Z_{\rm F}$  are the formal ionic charge of respective ions, r is their distance,  $A_2$  is a parameter characteristic of the *M*-F short-range interaction, and  $\alpha$  is a geometric constant.



FIG. 9. Structure of  $\beta$ -AlF<sub>3</sub>.



FIG. 10. Structure of NaMnCrF<sub>6</sub>.  $CrF_6$  and  $MnF_6$  share both edges and corners.

Furthermore, the minimization of the total energy provides a relation between  $A_2$ and the ionic charges:  $A_2 \propto Z_M Z_F$ . Thus, the harmonic force constant is proportional to  $Z_M Z_F$  and  $r^{-3}$ . The frequency of the SS vibration may be expressed by

$$\omega_{\rm SS}^2 = \frac{K \cdot e^2 \cdot Z_{\rm M} Z_{\rm F}}{m_{\rm F} r^3} \tag{2}$$

When neglecting the  $r^{-3}$  dependence  $\omega_{SS}$  is roughly proportional to  $Z_M^{1/2}$ . As previously outlined 520 cm<sup>-1</sup> is the order of magnitude of  $\omega_{SS}$  for  $F_{nb}$  vibrations in  $MF_6$  octahedra for  $M^{3+}$  ions ( $Z_M = 3$ ). Then, we can estimate  $\omega_{SS}$  as 600 and 420 cm<sup>-1</sup> for  $MF_6$  octahedra with  $M^{4+}$  and  $M^{2+}$  ions, respectively.

#### TABLE I

Observed Vibration Frequency and Mean *M*-F Distance (X-Ray Data)

Cation	Crystal	Frequency (cm <sup>-1</sup> )	<i>М</i> –F (Å)	Octahedral network
Zr <sup>4+</sup>	$Li_2ZrF_6(27)$	585	2.016	Isolated
	Cs <sub>2</sub> ZrF <sub>6</sub> (27)	577	2.112	Isolated
Ga <sup>3+</sup>	K2NaGaF6	558	1.940	Isolated
A13+	(NH4)3AIF6	546	1.900	Isolated
	K-AlF · H-O	536	1.908	One dim.
	Rb2AlF4 H2O	530	1.908	One dim.
	TI-AIF · H2O	516	1.908	One dim.
	Cs-NaAl <sub>3</sub> F <sub>12</sub>	536	1.801	Two dim.
	Na <sub>3</sub> Al <sub>3</sub> F <sub>14</sub>	534	1.790	Two dim.
	TIAIF	520	1.784	Two dim.
	KAIF	545	1.784	Two dim.
	B-RbAIF	528	1.808	Two dim.
Cr <sup>3+</sup>	Na <sub>5</sub> Cr <sub>3</sub> F <sub>14</sub>	544	1.840	Two dim.

In fact, as shown in Table I, strong  $A_{1g}$  bands are observed in the predicted frequency range for many compounds built with  $MF_6$  octahedra on  $M^{4+}$  and  $M^{3+}$  ions.

The last approximation allows us to identify one of the totally symmetric Raman lines as corresponding to SS of  $MF_6$  octahedra.

In order to go further we reported in Fig. 12  $\Gamma = \omega_{\rm SS} \cdot r^{3/2} \cdot Z_{\rm M}^{-1/2}$  as a function of the degree of connectivity between the octahedra. Figure 12 clearly shows that  $\omega_{ss}$ increases when the degree of connectivity decreases. This variation may be attributed to modifications of F-F short-range interactions. Effectively, in two-dimensionally connected octahedra the "SS" vibration involves only two nonbridging F<sub>nb</sub> atoms; on the other hand, in-phase motion of four  $F_{nh}$ atoms occurs in compounds with one-dimensionally connected octahedra. As illustrated in Fig. 13, the correlated motion of  $F_{nb}$  atoms enhances the F-F interatomic distance variations leading to an increase of the short-range F-F force constant. This effect is doubled in the case of isolated octahedra where 6  $F_{nb}$  atoms are moving simultaneously.

The motion of both  $M^{3+}$  and  $F^-$  ions in stretching mode should lead to higher frequencies: the reduced mass  $1/m_F$  in Eq. (2) is replaced by  $1/m_F + 1/m_{Cr}$  in the Rb<sub>2</sub>CrF<sub>5</sub> case; the group theory shows that the  $A_g$ modes  $(D_{2h})$  involve motion of both Cr<sup>3+</sup> and F<sup>-</sup> ions; the observed frequency (584 cm<sup>-1</sup>) is consistent with precedent remarks.

Apart from the SS mode involving nonbridging  $F_{nb}$  atoms (noted  $v^{S}$  in Almeida's work (28)), there exists a symmetric bending (SB) mode involving motion of bridging fluorine atoms  $F_{b}$  (noted  $\omega_{SS}$  in the same work) (Fig. 14). According to Almeida, the frequency  $\omega_{SB}$  of the SB vibration of  $F_{b}$ atoms is related to the frequency  $\omega_{SS}$  of  $F_{nb}$ atoms by

$$\omega_{\rm SB}^2 = 2 \cdot \omega_{\rm SS}^2 \cdot \cos^2 \theta/2. \tag{3}$$

This last expression has been derived using nearest-neighbor central forces between Mand F atoms; in fact, when plotting  $\omega$  versus cos  $\theta/2$  for various compounds (respectively  $Tl_2AlF_5 \cdot H_2O$ ,  $K_2AlF_5 \cdot H_2O$ ,  $Rb_2$ AlF<sub>5</sub>·H<sub>2</sub>O, Na<sub>5</sub>Al<sub>3</sub>F<sub>14</sub>,  $\beta$ -AlF<sub>3</sub>, and Cs<sub>2</sub>Na Al<sub>3</sub>F<sub>12</sub>, Fig. 15) with small variations of Al-F distances ( $\Delta r/r < 0.5\%$ ), it appears that



FIG. 11. Polarized Raman spectra of K<sub>2</sub>NaGaF<sub>6</sub> (a), Tl<sub>2</sub>AlF<sub>5</sub> · H<sub>2</sub>O (b), and Cs<sub>2</sub>NaAl<sub>3</sub>F<sub>12</sub> (c).



FIG. 11-Continued

the following law is much more realistic.

$$\omega_{\rm SB} = A + B \cdot \cos \theta/2, \qquad (4)$$

where A characterizes interactions other than M-F ones and B is estimated to be 514 cm<sup>-1</sup>. The difference between B and  $\omega_{\rm SS} \sqrt{2}$  (735 cm<sup>-1</sup>) gives an order of magnitude of the contribution of noncentral forces. It is noteworthy that A represents the symmetric bending frequency in the case of  $\theta = 180^{\circ}$ . This situation occurs in the high-temperature phase of RbAlF<sub>4</sub> where, according to Bulou (29), the corresponding  $X_7^2$  zone boundary mode is observed at 256 cm<sup>-1</sup> in agreement with the



FIG. 12. Variation of connecting scheme criterion  $\Gamma = r^{3/2}\omega/\sqrt{q}$  with degree of connectivity.



FIG. 13.  $F^-$  motion in SS  $A_{1g}$  mode for isolated octahedra, one-dimensional and bidimensional networks. When more  $F^-$  anions are involved, F-F short-range interactions increase leading to a higher symmetric stretching frequency  $\omega_{SS}$ .



FIG. 14. Schematic illustration of the symmetric bending SB vibration of bridging fluorine atom  $F_b$ .

232 cm<sup>-1</sup> extrapolated from our measurements.

(2) Case of tridimensional structures. All of the previously mentioned crystals display a symmetric stretching  $A_{1g}$  Raman line involving elongation of M-F<sub>nb</sub> bonds. The frequency of this SS mode depends mainly on the oxidation number of metal M and on the degree of connectivity of the  $MF_6$  octahedra. The interpretation of this particular line was justified by both lattice dynamics calculation and symmetry determination (25) of the Brillouin zone center normal mode with group theory. In the case of tridimensional networks (without  $F_{nb}$ ), each structure has to be seen as a special case. Figure 16 shows the Raman spectra of  $CoF_2$ ,  $\beta$ -AlF<sub>3</sub>, and NaMnCrF<sub>6</sub>:

(1) In the two AlF<sub>3</sub> forms, symmetry does not allow any Raman-active SS modes of M-F bonds; as expected, the characteristic SS line is not found.

(2) The rutile structure (MnF<sub>2</sub>, CoF<sub>2</sub>) is built with both corner and edge sharing of  $MF_6$  octahedra; nevertheless, as shown in Table II, the observed frequencies for  $A_{1g}$ mode are in the same range as the  $\omega_{SS}$ value.

(3) The NaMnCrF<sub>6</sub> spectrum shows three lines at 572, 538, and 358 cm<sup>-1</sup>; the first two lines are consistent with the existence of two different crystallographic sites for Cr<sup>3+</sup>. One Cr<sub>1</sub> (2d) participates in the  $A_{1g}$ vibration mode, giving a higher vibration at 572 cm<sup>-1</sup> (this case is to be compared to the case of Rb<sub>2</sub>CrF<sub>5</sub>); the other Cr<sub>2</sub> (1a) has no active Raman mode: the 538 cm<sup>-1</sup> line can be attributed to the  $\omega_{SS}$  vibration mode.

Since NaMnCrF<sub>6</sub> has a three-dimensional network involving Cr-F-Mn junctions, we could have expected the  $\omega_{SS}$  vibration to be perturbed by Mn<sup>2+</sup>; this effect



FIG. 15. Variation of symmetric bending mode frequency  $\omega_{SB}$  versus  $\cos \theta/2$ , where  $\theta$  is the M-F-M angle defined in Fig. 14.

TABLE II

Observed Vibration Frequency for the  $A_{1g}$ Mode and Mean M-F Distance (X-Ray Data)

Cation	Crystal	Frequency (cm <sup>-1</sup> )	<i>М</i> -F (Å)	
Cr <sup>3+</sup>	NaMnCrF <sub>6</sub>	538-572	1.909	
Mn <sup>2+</sup>	NaMnCrF <sub>6</sub> MnF <sub>2</sub>	358 338	2.130	
Co <sup>2+</sup>	CoF <sub>2</sub>	364	2.076	

is not observed. The 538 cm<sup>-1</sup> frequency very well agrees with values found in Table I. We explain this result by easily distorted  $MnF_6$  octahedra; the distortion is consistent with the known behavior of  $M^{2+}$  ions as connecting agents in fluoride glass systems.

The 358 cm<sup>-1</sup> line corresponds both to  $M^{II}$ -F ( $\omega_{SS}$ ) vibration and to  $M^{III}$ F<sub>6</sub> distortion (24, 25) because of the edge sharing of  $M^{II}$ F<sub>6</sub> and  $M^{III}$ F<sub>6</sub> octahedra. Table III compares observed and calculated frequencies of symmetric bending modes  $\omega_{SB}$ , using an expression derived from Eq. (3):

$$\omega_{\rm SB}^2 = [\omega_{\rm SS}^2(M^{\rm II}) + \omega_{\rm SS}^2(M^{\rm III})]/\cos^2\theta/2.$$
 (5)

A fairly good agreement is found for Mn– F–Cr bending where octahedra share corners (262/258), this is not the case when octahedra share edges (404/358): either Eq. (2) is not valid for this particular configuration or edge Mn–F–Cr vibration does not occur.

# TABLE III

Comparison of Observed and Calculated Frequencies of Symmetric Bending Modes  $\omega_{SB}$ 

Crystal	Bond	θ (°)	$\omega_{calc}$ (cm <sup>-1</sup> )	ω <sub>obs</sub> (cm <sup>-1</sup> )
$\frac{\text{NaMnCrF}_6}{\omega_{\text{Mn}} = 358 \text{ cm}^{-1}}$	Mn-F-Cr	132.1	262	258
$\omega_{\rm Cr} = 538 \ \rm cm^{-1}$	Mn F Cr	103.1 102.2	404	358

# III. Raman Data on Disordered Phases or Glasses

#### 1. The RbNiCrF<sub>6</sub>-type Pyrochlores

In the case of pyrochlore structure (CsZn GaF<sub>6</sub>, (30)), all fluorine atoms are bounded either to ZnF<sub>6</sub> or GaF<sub>6</sub> octahedra (statistically disordered—Figs. 17 and 18, Table IV). Application of group theory shows that in the Fd3m space group, vibrations along the M-F bond are forbidden.

Two interpretations of the spectrum are possible, but both are inconsistent with the space group Fd3m:

—The statistical partition of Zn and Ga on the same site must result in many Zn–F– Ga connections in contradiction with the higher value found at 604 cm<sup>-1</sup> (broad line) compared to 538 cm<sup>-1</sup> for  $\omega_{Cr}$  in NaMn CrF<sub>6</sub>. The broadening of the line could indicate both the cationic disorder and a notable amount of Ga–F–Ga connections.

—The 604 cm<sup>-1</sup> vibration is a  $\omega_{SS}$  vibration indicating that the *Fd3m* space group is wrong as suggested by a previous EXAFS study (31). The lack of translational symmetry modifies the selection rules, allowing activity for normally forbidden modes.

# 2. Three-Dimensional Transition Metal Fluoride Glasses

The "statistical" cationic distribution in the 16c site of  $CsZnGaF_6$  pyrochlore structure is probably a situation not far from that which exists in 3d transitional element

TABLE IV Comparison of Observed and Calculated Frequencies for Bending Modes

Crystal	Bond	θ (°)	$\omega_{calc}$ (cm <sup>-1</sup> )	ω <sub>obs</sub> (cm <sup>-1</sup> )
CsZnGaF6	Ga-F-Ga	149.1	369	385
$\omega_{Ga} = 530 \text{ cm}^{-1a}$ $\omega_{Zn} = 360 \text{ cm}^{-1}$	Ga-F-Zn Zn-F-Zn	137.8 129.5	231 220	210

<sup>a</sup> Estimated from  $r^{3/2} \cdot \omega = f(\sqrt{Z_m})$  curve.

fluoride glasses. Their structure can be described in two ways:

 $-Mt^{II}F_6$  and  $Mt^{III}F_6$  octahedra are linked

together essentially by corners but the

 $Mt^{II}F_6$  octahedra are more connected than the  $Mt^{III}F_6$  octahedra; large cations (Pb<sup>2+</sup>, Ba<sup>2+</sup>...) occupy the large holes of the tridimensional octahedral network very close to that of lacunary ReO<sub>3</sub> type.



FIG. 16. Polarized spectra of  $CoF_2$ ,  $\beta$ -AlF<sub>3</sub>, and NaMnCrF<sub>6</sub>.



—Three-dimensional transition metals fit the octahedral sites of a quasi-compact fluoride ion packing; the perturbation in the network is due to the insertion of the large cations ( $Pb^{2+}$ ,  $Ba^{2+}$ ..., same size as  $F^{-}$ ) in the  $F^{-}$  packing.



FIG. 17. Structure of RbNiCrF<sub>6</sub>-type pyrochlore.

Figure 19 shows the average structure of TMFG.

Bose-Einstein (32) corrected Raman spectra of TMFG containing only  $M^{III}$  ions (Glass A) or  $M^{II}$  ions (Glass B) and both  $M^{II}$  and  $M^{III}$  ions (Glasses C and D) in sixfold coordination (from previous structural studies) are shown in Fig. 20. Glasses A, C, and D exhibit a narrow V-V polarized line in the range 500-600 cm<sup>-1</sup>:

—The replacement of  $MnF_2$  with  $ZnF_2$ (Glasses C, D) does not affect the 560 cm<sup>-1</sup> line leading us to the conclusion that this line is due to  $M^{III}$ —F stretching.

--We tried to compare the width (HWHM) of both this Raman line and the EXAFS radial distribution functions (RDF, (6)) in the case of the same glass (Glass D). Assuming again that the  $\omega_{SS}$  frequency can be written as: constant  $\cdot r^{-3/2}$ , we can see that  $\Delta\omega/\omega = 3/2 \Delta r/r$ ; this leads to  $\Delta r = 0.085$  Å (very close to the EXAFS value  $\Delta r = 0.08$  Å).



FIG. 18. Polarized Raman spectrum of CsZnGaF<sub>6</sub>.

—This well-defined line, attributed to  $M^{\rm III}$  cations, when associated with the broadened contributions of Pb<sup>2+</sup> and  $M^{2+}$  in the low frequency range confirms that in these glasses the short-range order depends upon  $M^{\rm III}$ F<sub>6</sub> octahedra, while Pb<sup>2+</sup> and  $M^{2+}$  could have the classical modifier behavior.

The comparison of variation of connecting scheme dependent criterion  $\Gamma = r^{3/2} \cdot \omega/\sqrt{q}$  with the glass composition (Table V) leads to conclusions, consistent with the known structural models:

-Glass A ( $\Delta r = 0.086$  Å). The molar composition ratio In/(Pb + Ba)  $\approx 1$  implies the predominance of InF<sub>6</sub> species and the possible existence of In-F-In connections; the  $\Gamma$  value is in favor of a monodimensional network.

-Glass B ( $\Delta r = 0.16$  Å). This glass (more unstable), contains only  $M^{11}F_6$  enti-



FIG. 19. Structural model for TMFG network:  $\bullet$ ,  $M^{2+}$ ;  $\bigcirc$ ,  $M^{3+}$ .

TABLE V Variation of  $\Gamma = r^{3/2} \cdot \omega/\sqrt{q}$  with Glass Composition

Glass	Composition	$\omega_{SS}$ (cm <sup>-1</sup> )	<i>r(M</i> -F) (Å)	Г
A	19.0PbF2-23.8BaF2-47.6InF3	500	2.02	829
	1.9AlF <sub>3</sub> -4.9SrF <sub>2</sub> -2.8YF <sub>3</sub>			
В	$20BaF_2 - 50ZnF_2 - 30YF_3$	410	2.01 <sup>a</sup>	826
С	35.3PbF <sub>2</sub> -23.5MnF <sub>2</sub> -34.3GaF <sub>3</sub> 2.0AlF <sub>3</sub> -4.9YF <sub>3</sub>	560	1.90ª	847
D	35.3PbF <sub>2</sub> -23.5ZnF <sub>2</sub> -34.3GaF <sub>3</sub> 2.0AlF <sub>3</sub> -4.9YF <sub>3</sub>	560	1.90 <sup>a</sup>	847

<sup>a</sup> EXAFS results.

ties; the  $\omega_{ss}$  line found at 410 cm<sup>-1</sup> is large, indicative of the octahedra's distortion. The value of Zn/(Ba + Y), related to the  $\Gamma$ value, predicts a situation analogous to that of Glass A. --Glass C, D ( $\Delta r = 0.085$  Å). According to the lower value of ratio Ga/(Pb +  $M^{II}$ ), it can be expected that Ga-F- $M^{II}$  connections become predominant; the  $\Gamma$  value indicates mostly isolated or "pseudo"-iso-



FIG. 20. V-V and V-H Bose-Einstein corrected spectra of glasses A, B, C, and D. (For glass composition refer to Table V.)



lated  $M^{III}F_6$  octahedra. We already noticed tent with the model of alternated Ga- $M^{II}$ 

that in NaMnCrF<sub>6</sub>, Mn has no effect on  $M^{III}$ -F  $\omega_{SS}$  vibration. This view is consis- we cannot exclude the existence of the

Ga-F-Ga connection since the 604  $cm^{-1}$  frequency found in the pyrochlore is similar to that value.

# **IV. Conclusion**

Raman spectra recording of small crystals (down to 0.1 mm) was made possible thanks to a micro-Raman spectrometer. Study of single crystals of octahedrally coordinated fluorides has shown that the symmetric stretching frequency  $\omega_{SS}$  mainly depends on the oxidation number of the metal and then on the degree of connectivity; but the case of three-dimensional structures is more complex.

TMFG study shows unambiguously that the less-distorted octahedra are  $M^{III}F_6$ , with predominance of the  $\omega_{SS}$  vibration of  $M^{III}$ -F bonds. The short-range order is mainly due to trivalent metal octahedra; the structural view is consistent with that given previously by other studies.

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